Crystal growth, transport properties, and crystal structure of the single-crystal La\textsubscript{2-x}Ba\textsubscript{x}CuO\textsubscript{4}(x=0.11)

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Crystal growth, transport properties, and crystal structure of the single-crystal \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x = 0.11) \)

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We have attempted the crystal growth by the traveling-solvent floating-zone (TSFZ) method of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x \approx 1/8) \), where the superconductivity is strongly suppressed. Under flowing \( \text{O}_2 \) gas at high pressure (4 bars), we have succeeded in growing single crystals for \( x = 0.11 \) of 5 mm in diameter and 20 mm in length. Both in-plane and out-of-plane electrical resistivities of single-crystal \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x = 0.11) \) exhibit a clear jump at \( \approx 53 \) K. The temperature corresponds to the structural phase transition temperature between the orthorhombic midtemperature and tetragonal low-temperature phases, \( T_{d2} \). Both in-plane thermoelectric power and Hall coefficient drop rapidly at \( T_{d2} \) with decreasing temperature. These results are quite similar to those observed in single-crystal \( \text{La}_{1.6-x}\text{Nd}_{x}\text{Sr}_x\text{CuO}_4 \) \( (x \approx 1/8) \), suggesting that the so-called static stripe order of holes and spins in the \( \text{CuO}_2 \) plane is formed below \( T_{d2} \) in \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x \approx 1/8) \) as well as in \( \text{La}_{1.6-x}\text{Nd}_{x}\text{Sr}_x\text{CuO}_4 \) \( (x \approx 1/8) \).

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I. INTRODUCTION

Since the discovery of the anomalous suppression of superconductivity in \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x \approx 1/8) \),\(^{1,2}\) the so-called \( 1/8 \) anomaly has been a subject of considerable research attention. In recent years, the \( 1/8 \) anomaly has been found not only in a series of \( \text{La}_{2-1} \) high-\( T_c \) superconductors\(^{3,4}\) but also in the Bi-2212\(^{5,6}\) and Y-123 ones\(^{7-9}\), which means that the \( 1/8 \) anomaly is common to high-\( T_c \) superconductors,\(^{18-21}\) are pinned by the TLT structure, leading to the appearance of the static stripe order and the suppression of superconductivity.\(^{26}\) Accordingly, the preparation of the high-quality single crystals of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x \approx 1/8) \) and the detailed study on the possible stripe order are of much importance.

Several attempts to grow single crystals of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) by the traveling-solvent floating-zone (TSFZ) method have been reported so far.\(^{25-30}\) However, no single crystal of good quality with \( x \approx 1/8 \) has been grown in flowing \( \text{O}_2 \) gas at pressures below \( \approx 2 \) bars. Therefore, we have attempted to grow single crystals of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x \approx 1/8) \) by the TSFZ method in flowing \( \text{O}_2 \) gas at high pressures, based on the analogy of the successful growth of \( \text{La}_{2-x}\text{Sr}_x\text{CuO}_4 \) single crystals.\(^{31-35}\) Then, we have investigated the transport properties of the obtained single crystals.

II. EXPERIMENTAL DETAILS

In the crystal growth of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) \( (x \approx 1/8) \), \( \text{La}_{2}\text{O}_3 \), \( \text{BaCo}_3 \), and \( \text{CuO} \) powders were used as raw materials of the feed rod and the solvent. For the feed rod, the powders in the molar ratio of \( \text{La}:\text{Ba}:\text{Cu} = 1.875:0.125:1 \) were mixed and pre-fired in air at 900 °C for 12 h. After pulverization, the pre-fired materials were mixed and sintered in air at 1100 °C for 24 h. This process of mixing and sintering was repeated four times to obtain homogeneous powders of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \). Next, 1 mol % \( \text{CuO} \) powders were added to the powders of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \) and fired in air at 900 °C for 12 h. After pulverization, the pre-fired powders were mixed with the solvent and pre-fired in air at 900 °C for 12 h. After pulverization, the pre-fired materials were mixed and sintered in air at 1100 °C for 24 h. This process of mixing and sintering was repeated four times to obtain homogeneous powders of \( \text{La}_{2-x}\text{Ba}_x\text{CuO}_4 \). Next, 1 mol % \( \text{CuO} \) powders were added to the powders of...
La$_{2-x}$Ba$_x$CuO$_4$ and mixed thoroughly, in order to obtain tightly sintered feed rods in the sintering process and also to compensate for evaporated CuO in the TSFZ growth process. The obtained fine powders were put into thin-walled rubber tubes and formed into cylindrical rods under a hydrostatic pressure of 2.4 kbars. The typical dimensions of the rods were 6 mm in diameter and 120 mm in length. In the growth process of La$_{2-x}$Ba$_x$CuO$_4$, one of the most serious problems is deep penetration of the molten zone into the feed rod, which makes the molten zone unstable. To avoid such a situation, it is important for the feed rod to be sintered as tightly as possible. Therefore, we measured the melting temperature of La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.125$), and the final sintering was carried out at 1250 °C just below the melting temperature for 24 h in air.

For the solvent rod, the composition of the raw materials was richer in Cu; typically La$_{0.875}$Ba$_{0.125}$;Cu=3:7 in the molar ratio. Powders of the raw materials were mixed and prefired in air at 900 °C for 12 h. After pulverization, the prefired materials were mixed and formed into cylindrical rods. Then, the final sintering was performed in air at 900 °C for 12 h. The sintered rods were sliced in pieces and a piece of ~0.4 g was used as a solvent for the TSFZ growth.

The TSFZ growth was carried out in an infrared heating furnace equipped with a quartet ellipsoidal mirror (Crystal Systems Inc., Model FZ-T-4000-H). Under flowing O$_2$ gas at high pressure (4 bars), the zone traveling rate was 1.0 mm/h and the rotation speed of the feed rod and the grown crystal was 30 rpm in the opposite direction.

In order to fill up oxygen vacancies and to remove the strain, the as-grown crystals were postannealed in flowing O$_2$ gas at 1 bar and 900 °C for 50 h, cooled down to 500 °C at a rate of 8 °C/h, kept at 500 °C for 50 h, and then cooled down to room temperature at a rate of 8 °C/h.

The dc magnetic susceptibility was measured with a superconducting quantum interference device magnetometer (Quantum Design, Model MPMS-XL5). Electrical resistivity measurements were carried out by the dc four-probe method. The thermoelectric power was measured by the dc method with a temperature gradient of ~0.5 K across a crystal. The Hall coefficient was measured by the ac method with a frequency of 30 Hz. Powder x-ray diffraction measurements were also performed in a temperature range between 10 K and 280 K, in order to estimate the structural phase transition temperatures.

III. EXPERIMENTAL RESULTS

A. Characterization of grown crystals

We succeeded in keeping the molten zone stable under flowing O$_2$ gas at 4 bars during the TSFZ growth. An as-grown crystal is shown in Fig. 1. The dimensions were 5 mm in diameter and 70 mm in length. A few days later, however, the crystal of a half or over was broken into pieces in air, because the initially grown portion of the crystal contained some inclusions such as La$_2$O$_3$, which was confirmed by the powder x-ray diffraction measurements.$^{36}$ A single crystal of 5 mm in diameter and 20 mm in length was obtained from the part grown in the last stage.

Details of the domain structure of the grown crystal were investigated by the x-ray back-Laue photography. As shown in Fig. 2, fourfold symmetric spots were clearly observed in the photograph of a surface parallel to the growth direction, indicative of the crystal as a single crystal. Moreover, the broadness of the spots was similar to that in La$_2$-Sr,CuO$_4$ single crystals.$^{35}$ We also checked the crystal by the powder x-ray diffraction. There could be seen Bragg peaks of the La-214 phase and no impurities such as La$_1$+$_x$Ba$_{2-x}$Cu$_2$O$_7$-, La$_4$BaCu$_5$O$_{15}$, and La$_2$O$_3$ reported in some earlier papers.$^{27,28,30}$ Accordingly, it is concluded that the quality of the grown single crystal is good.

The Ba content of the grown crystal was estimated by the inductively coupled plasma atomic-emission-spectrometry (ICP-AES) to be $x=0.11$, which is a little smaller than that of the feed rod. It may be due to some evaporation of Ba and also due to concentration of Ba into the molten zone in the TSFZ growth process. In fact, the Ba content in the molten zone, estimated by the ICP-AES after the growth, was La:Ba = 0.94:1.06 in the molar ratio. Therefore, it is considered that the analyzed ratio of the solvent of La:Ba = 0.94:1.06 is more appropriate than the starting composition of La:Ba = 1.875:0.125 for the crystal growth of $x=0.11$. This content
of Ba is much larger than the Sr content in the molten zone in the case of the growth of the overdoped La$_{2-x}$Sr$_x$Cu$_{1-y}$Zn$_y$O$_4$; typically La: Sr = 1.5:0.5 in the molar ratio. The Cu content in the molten zone after the growth was estimated by the ICP-AES to be La$_{0.94}$Ba$_{1.06}$:Cu = 2.0:8.0 in the molar ratio, which is richer than that in the starting composition. It has been reported, on the other hand, that the starting composition of the solvent (La, Ba):Cu = 2:8 results in the inclusion of (La, Ba)$_2$CuO$_3$ in the grown crystals. Therefore, the starting composition of the solvent (La, Ba):Cu = 3:7 may be quite suitable for the crystal growth of $x = 0.11$.

The oxygen content of the postannealed crystal of La$_{1.89}$Ba$_{0.11}$CuO$_4$ was checked by iodometry. As a result, there was no oxygen deficiency within the experimental accuracy. That is, $\delta = 0.00 \pm 0.01$ for La$_{2-x}$Ba$_x$CuO$_4$.37 indicating that the postannealing process mentioned in Sec. II is enough to obtain stoichiometric single crystals of La$_{2-x}$Ba$_x$CuO$_4$.

Figure 3 shows the temperature dependence of the zero-field-cooled (ZFC) and field-cooled (FC) dc magnetic susceptibilities of La$_{1.89}$Ba$_{0.11}$CuO$_4$ ($x = 0.11$). The solid lines guide the reader’s eye.

**FIG. 3.** Temperature dependence of the zero-field-cooled (ZFC) and field-cooled (FC) dc magnetic susceptibilities of La$_{2-x}$Ba$_x$CuO$_4$ ($x = 0.11$). The solid lines guide the reader’s eye.

**B. Transport properties and crystal structure**

Figure 4 shows the temperature dependence of the in-plane ($\rho_{ab}$) and out-of-plane ($\rho_c$) electrical resistivities in the single-crystal La$_{2-x}$Ba$_x$CuO$_4$ ($x = 0.11$). The $\rho_{ab}$ exhibits a metallic behavior in the normal state. The residual resistivity, defined as the extrapolated value of the resistivity in the normal state to $T = 0$ K, is as small as $\sim 10^{-5}$ $\Omega$ cm, which is similar to those of high-quality single crystals of La$_{2-x}$Sr$_x$CuO$_4$.33,38 The $\rho_c$ exhibits a semiconducting behavior in the normal state. A kink is observed at 256 K, corresponding to the structural phase transition temperature between the tetragonal high-temperature (THT, space group: I4/mmm) and OMT phases, $T_{d1}$. The anisotropy, $\rho_c/\rho_{ab}$, is $\sim 10^3$ at room temperature, which is almost the same as that of La$_{2-x}$Sr$_x$CuO$_4$ ($x \sim 1/8$).38 The value of $T_c$, defined as the midpoint of the superconducting transition curve, is 17.8 K.39 The values of $T_c$ and $T_{d1}$ are in good agreement with those of the polycrystalline sample with $x = 0.11$, respectively, as shown in Fig. 5. A clear jump is observed at $\sim 53$ K in the temperature dependence of both $\rho_{ab}$ and $\rho_c$, as shown in the inset of Fig. 4, though no jump has been observed in the polycrystalline samples with $x \sim 1/8$. The jumps are considered to be due to the structural phase transition between the OMT and TLT phases, as in the case of the single-crystal La$_{1.6-x}$Nd$_{0.4}$Sr$_x$CuO$_4$ ($x = 0.12$).22 In fact, the temperature roughly corresponds to $T_{d2}$ of the polycrystalline sample with $x = 0.11$ as also shown in Fig. 5. These results on the electrical resistivity also suggest that the composition of the single crystal is almost the same as that of the polycrystalline sample with $x = 0.11$ and that the quality of the single crystal is quite good.

The structural phase transition temperature $T_{d1}$ and $T_{d2}$ of the single-crystal La$_{2-x}$Ba$_x$CuO$_4$ ($x = 0.11$) have actually been estimated from the powder x-ray diffraction measurements at low temperatures. Figure 6(a) is a stack plot showing the temperature dependence of the diffraction profile of the (220)$_{THT}$ reflection in the notation of the THT phase, which is sensitive to the tetragonal-orthorhombic phase transitions. A single peak is observed at high temperatures above $T_{d1}$. The peak splits progressively with decreas-
cause of the second-order transition at $T_d$. FWHM’s increase gradually with decreasing temperature, as shown in Figs. 6 and 7, together with the $\rho_{ab}$ vs $T$ plot. The $R_H$ was measured in a magnetic field of 4 T. The $S_{ab}$ drops rapidly at $T_{d_1}$ and decreases below $T_{d_2}$ with decreasing temperature. Moreover, it changes the sign somewhat at low temperatures below $\sim 25$ K, as shown in the inset of Fig. 7(b), which is well known to be characteristic of the 1/8 anomaly. The $R_H$ also drops rapidly at $T_{d_2}$ and decreases below $T_{d_2}$ with decreasing temperature. Then, a conspicuous reversal in the sign of $R_H$ is observed at low temperatures below $\sim 25$ K, where $S_{ab}$ also exhibits the sign reversal. These anomalous behaviors of $S_{ab}$ and $R_H$ are similar to those observed in La$_{1.6-x}$Nd$_{0.4}$Sr$_x$CuO$_4$ ($x=0.12$), though the sign reversal of $R_H$ is not well defined in La$_{1.6-x}$Nd$_{0.4}$Sr$_x$CuO$_4$.22

IV. DISCUSSION

We first discuss the clear jump or drop at $T_{d_2}$ in the temperature dependence of $\rho_{ab}$, $\rho_{c}$, $S_{ab}$, and $R_H$, which we have observed in La$_{2-x}$Ba$_x$CuO$_4$. These clear transport anomalies at $T_{d_2}$ have been formerly observed in La$_{1.6-x}$Nd$_{0.4}$Sr$_x$CuO$_4$ ($x\sim 1/8$).22 However, this is an important observation of such anomalies in “La$_{2-x}$Ba$_x$CuO$_4”.43

This is analogous to that observed in the polycrystalline La$_{2-x}$Ba$_x$CuO$_4$. This is analogous to that observed in the polycrystalline La$_{2-x}$Ba$_x$CuO$_4$.22 Thus, it is possible that the change of charge dynamics relevant to the change of lattice occurs at $T_{d_2}$.

In general, signs of the thermoelectric power and the Hall coefficient reflect the sign of carriers. Therefore, it is very likely that the observed sign reversals in the temperature dependences of $S_{ab}$ and $R_H$, as shown in Fig. 7, have the same origin. The sign reversal of $S_{ab}$ in the single-crystal La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.11$) is very small, compared with that in the polycrystalline La$_{2-x}$Ba$_x$CuO$_4$ ($x=1/8$).13 It may be due to the smaller value of $x$ than 1/8 where the sign reversal is the most conspicuous in the polycrystalline La$_{2-x}$Ba$_x$CuO$_4$. As for the sign reversal of $R_H$, it is often observed in the superconducting fluctuation regime, though it has not yet been understood clearly. It is an empirical fact that the sign reversal of $R_H$ has a strong magnetic-field dependence in the superconducting fluctuation regime. That is, the magnitude of the sign reversal of $R_H$ decreases with increasing magnetic field, accompanied by the broadening of the superconducting transition curve in resistivity.44 Figure 8 displays the temperature dependence of $R_H$ in various magnetic fields, measured using another batch of the single-
crystal La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.11$). It is found that the magnitude of the sign reversal of $R_H$ increases with increasing magnetic field and saturates in higher fields above 6 T. Moreover, the sign-reversal temperature is independent of the onset temperature of the superconducting transition $T_{c_{onset}}$ estimated from the resistivity in each magnetic field. Therefore, attributing the observed reversal in the sign of $R_H$ in the single-crystal La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.11$) to the superconducting fluctuation is difficult. The origin of the sign reversals in $S_{ab}$ and $R_H$ is not clarified, but these behaviors are considered as characteristic of the 1/8 anomaly in a series of La-214 systems.

We now discuss the experimental results in the single-crystal La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.11$) from the viewpoint of the static stripe order. The observed anomalous behaviors in $\rho_{ab}$, $\rho_c$, $S_{ab}$, and $R_H$ are quite similar to those observed in La$_{1.6-x}$Nd$_{0.4}$Sr$_x$CuO$_4$ ($x=1/8$) where the static stripe order is formed at low temperatures below $T_d$. In particular, Noda et al. have suggested for La$_{1.4-x}$Nd$_{0.6}$Sr$_x$CuO$_4$ that the rapid decrease in $R_H$ below $T_d$ is a typical behavior of the one-dimensional transport associated with the formation of the stripe order. Therefore, we are much convinced that the static stripe order of holes and spins is formed at low temperatures below $T_d$ also in La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.11$), though it was already predicted from the experimental results in the polycrystalline samples. Although $\rho_{ab}$ exhibits an upturn at low temperatures below $T_d$ in La$_{1.6-x}$Nd$_{0.4}$Sr$_x$CuO$_4$ ($x=0.12$), $\rho_{ab}$ in the single-crystal La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.11$) does not exhibit any upturn but exhibits a metallic behavior. It is because the stripe order may be highly fluctuating at low temperatures below $T_d$ in spite of the pinning by the TLT structure, as in the case of La$_{1.6-x}$Nd$_{0.4}$Sr$_x$CuO$_4$ ($x=0.10$). To be sure of the above speculation, a direct observation of the static stripe order from the neutron scattering experiment is now being planned.

V. SUMMARY

We have succeeded in growing high-quality single crystals of La$_{2-x}$Ba$_x$CuO$_4$ ($x=0.11$) by the TSFZ method under
flowing O\textsubscript{2} gas at high pressure (4 bars). We have investigated the temperature dependence of $\rho_{ab}$, $\rho_c$, $S_{ab}$ and $R_H$ and the crystal structure from the powder x-ray diffraction measurements at low temperatures. We have observed a clear jump in both $\rho_{ab}$ and $\rho_c$ at $T_d$. We have also found that both $S_{ab}$ and $R_H$ drop rapidly at $T_d$ and decrease below $T_d$ with decreasing temperature. These anomalous behaviors are analogous to, but much clearer than, those observed in the decreasing temperature. These anomalous behaviors are gated the temperature dependence of $r$.


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36 It is well known that La$_2$O$_3$ changes to La(OH)$_3$ by absorbing water in air, leading to the destruction of the grown crystal.
37 S. Watauchi (private communications).
40 The $T_c$ value estimated from the resistivity is higher than that.
from the susceptibility, which is also known in polycrystalline sample with \( x \approx \frac{1}{8} \). The phase of a little lower concentration of Ba than \( x = 0.11 \) remaining in the filamentary shape may exhibit superconductivity at temperatures higher than 10.2 K.


43 No anomaly has been observed in the transport properties of the single-crystal \( \text{La}_{2-x}\text{Ba}_{x}\text{CuO}_4 (x = 0.095) \), which may be due to the value of \( x \) being far from 1/8. Y. Abe, Y. Ando, J. Takeya, H. Tanabe, S. Watauchi, I. Tanaka, and H. Kojima, Phys. Rev. B 59, 14753 (1999).


